ON STRESS-STRAIN RELATIONS FOR ISOTROPIC RIGID PERFECTLY PLASTIC SOLIDS*

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Abstract. Sufficient conditions under which principal directions of stress and strain rate must coincide are established rigorously. It is the coincidence of these directions which permits a proper interpretation of principal strain rate components in principal stress space.

Introduction. A rigid perfectly plastic solid is characterized by a "yield" or "limit" surface $f(\sigma_{ij}) = 0$ in stress component space which encloses allowable stress states. The strain rate tensor $\dot{\epsilon}_{ij}$ must vanish if σ_{ij} is inside the surface and, if strain rate component coordinates are superposed upon stress component coordinates, must be normal to the surface if σ_{ij} is on the surface.

If the yield surface is that of Mises (that is, if $f(\sigma'_{ij}) = \frac{1}{2}\sigma'_{ij}\sigma'_{ij}$, where σ'_{ij} is the stress deviator tensor) then normality requires that for some scalar λ , $\dot{\epsilon}_{ij} = \lambda(\partial f/\partial \sigma_{ij}) = \lambda \sigma'_{ij}$; coincidence of principal directions of stress and strain rate is then immediate since the principal axes of the stress and stress deviator tensors coincide.

More generally, if f is a function of stress invariants which is differentiable in the components of σ_{ij} then the principal directions of σ_{ij} and $\partial f/\partial \sigma_{ij}$ conicide; by normality, then, the axes of σ_{ij} and $\dot{\epsilon}_{ij}$ coincide. Alternatively, a stress-strain rate equation of the form $\dot{\mathbf{c}} = \mathbf{f}(\mathbf{d})$ can be postulated and the material taken to be isotropic if and only if for all orthogonal transformations s the equation s $\dot{\mathbf{c}}$ $s^{-1} = \lambda \mathbf{f}(s \sigma s^{-1})$ holds, where s^{-1} is the inverse of s. From this definition of "response symmetry", coincidence follows from a well-known theorem [1].

Hill [2] and Shield [3], among others, have taken coincidence for the rigid perfectly plastic Tresca material on the premise of material isotropy but did not supply a proof. Neither of the results quoted above are applicable since the first requires a differentiable yield function and the second a unique correspondence between the tensors of stress and strain rate or their deviators. Moreover, coincidence does not hold for an isotropic frictional material [4], which obeys the verbal hypotheses upon which coincidence for the Tresca material is generally taken.

In Sec. 1 a pictorial description of the rigid perfectly plastic Tresca material is given and it is explained why coincidence does not hold for the conceptually similar frictional material. In Sec. 2 it is shown that coincidence must hold for a rigid perfectly plastic material if the question of "whether or not a stress state represents yield" is to be independent of the relative orientation of material and principal stress directions.

1. The Tresca and frictional materials. The relation between the tensors of stress and strain rate for the Tresca material is explained in Fig. 1(a) if the principal stress components are ordered as $\sigma_1 > \sigma_2 > \sigma_3$. Deformation occurs by simple shearing in the direction of the shear stress vector **T** if the magnitude τ of **T** reaches a critical value k. Since the planes of maximum shearing, called slip planes, are orthogonal, the simple

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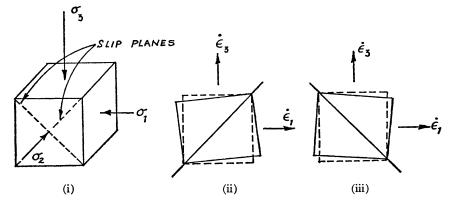


Fig. 1(a). Rigid perfectly plastic Tresca material

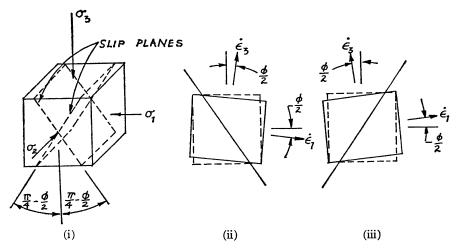


Fig. 1(b). Frictional material; φ :angle of internal friction

FIGURE 1. Configuration of a block before and after a small homogeneous deformation. Displacements are chosen so that block center remains stationary and one of the sets of zero extension does not rotate.

shearing occurs simultaneously on both sets of slip planes; the first order difference in the deformed configurations of Figs. (ii) and (iii) is in rigid rotation only.

Should two principal stresses be equal, say $\sigma_1 = \sigma_2$, then τ reaches the value k simultaneously on all planes with normal inclined at $\pi/4$ to the σ_3 direction; deformation can occur by simple shearing along any of these mutually perpendicular sets of planes. Deformations other than plane are achieved by successive simple shears on different mutually perpendicular sets of planes. The strain rate tensor is then taken as a suitable time average of the successive simple shears. The directions of σ_3 and $\dot{\epsilon}_3$ thus remain coincident.

The relation between the tensors of stress and strain rate for plane motions of a frictional material is explained in Fig. 1(b). Deformation occurs once again by simple shearing on a slip plane but slip planes and planes of maximum shear stress do not coincide. Neither set has a material preference for shearing, so the principal axes of strain rate

can be inclined either as in Fig. (ii) or in Fig. (iii). For this material, strain rate is not normal to the limit surface.

2. Coincidence for isotropic rigid perfectly plastic materials. No regularity assumptions on the yield surface are necessary for the following proof of coincidence. The notions of isotropy and normality will not be made precise; rather, necessary conditions for their satisfaction are stated.

A seemingly reasonable necessary condition for a perfectly plastic material to be isotropic is the following: if σ_{ij}^0 is an arbitrary point on a yield surface, then σ_{ij} is on the surface also if σ_{ij} differs from σ_{ij}^0 in principal directions but not principal values.

Say that at a point on a hypersurface there are one or more hypercurves on the surface with uniquely determined tangent directions. The hypersurface itself need not be regular there. A seemingly reasonable necessary condition for a vector to be "normal" to the hypersurface there is that the inner product of the vector with uniquely determined tangent directions should vanish.

Without loss of generality, take x_1 , x_2 , and x_3 axes to coincide respectively with the directions of σ_1^0 , σ_2^0 , and σ_3^0 as shown in Fig. 2. The hypercurve defined by

$$||\sigma_{ij}|| = \begin{bmatrix} \sigma_{11} & \sigma_{12} & \sigma_{13} \\ \sigma_{12} & \sigma_{22} & \sigma_{23} \\ \sigma_{13} & \sigma_{23} & \sigma_{33} \end{bmatrix} = \begin{bmatrix} \sigma_{3}^{0} + (\sigma_{1}^{0} - \sigma_{3}^{0}) \cos^{2} \alpha & 0 & (\sigma_{1}^{0} - \sigma_{3}^{0}) \cos \alpha \sin \alpha \\ 0 & \sigma_{2}^{0} & 0 \\ (\sigma_{1}^{0} - \sigma_{3}^{0}) \cos \alpha \sin \alpha & 0 & \sigma_{1}^{0} - (\sigma_{1}^{0} - \sigma_{3}^{0}) \cos^{2} \alpha \end{bmatrix}$$
(1)

is generated by the rotation of principal directions of Fig. 2 and thus lies on the yield

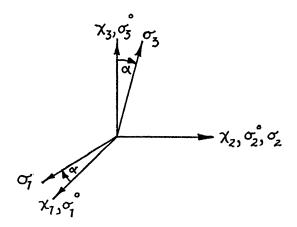


Figure 2. Principal stress directions in physical body before and after rotation about x_2 axis.

surface. The direction of the tangent vector to the hypercurve is given uniquely by $d\sigma_{ij}/d\alpha$; thus if the strain rate tensor $\dot{\epsilon}_{ij}^0$ at σ_{ij}^0 is normal to the yield surface, then $(d\sigma_{ij}/d\alpha)\dot{\epsilon}_{ij}^0 = 0$ there.

However, it follows from differentiating Eq. (1) that

$$\left| \left| \frac{d\sigma_{ii}}{d\alpha} \right| \right| = \begin{bmatrix} -2(\sigma_1^0 - \sigma_3^0) \cos \alpha \sin \alpha & 0 & (\sigma_1^0 - \sigma_3^0)(\cos^2 \alpha - \sin^2 \alpha) \\ 0 & 0 & 0 \\ (\sigma_1^0 - \sigma_3^0)(\cos^2 \alpha - \sin^2 \alpha) & 0 & 2(\sigma_1^0 - \sigma_3^0) \cos \alpha \sin \alpha \end{bmatrix}$$

and since $\alpha = 0$ at $\sigma_{ij} = \sigma_{ij}^0$ that

$$(d\sigma_{ij}/d\alpha)\dot{\epsilon}_{ij}^0 = 2(\sigma_1^0 - \sigma_3^0)\dot{\epsilon}_{13}^0.$$

If $\dot{\epsilon}_{13} \neq 0$, a contradiction has thus been attained provided $\sigma_1^0 \neq \sigma_3^0$. Similarly, by considering hypercurves consisting of stress points σ_{ij} which are rotated versions of σ_{ij}^0 about the σ_1^0 and σ_3^0 axes, the possibilities of $\dot{\epsilon}_{12} \neq 0$ and $\dot{\epsilon}_{23} \neq 0$ can be ruled out provided respectively that $\sigma_1^0 \neq \sigma_2^0$ and $\sigma_2^0 \neq \sigma_3^0$. Thus, as was shown pictorially for the Tresca material in the previous section, the principal axes of stress and strain rate coincide if all three principal stress components are different.

In the degenerate case of equality of two principal stress components, say $\sigma_2^0 = \sigma_3^0$, then $\dot{\epsilon}_{12}^0 = \dot{\epsilon}_{13}^0 = 0$ follows as before. Thus, the direction of σ_1^0 is also a direction of principal strain rate and the directions of $\dot{\epsilon}_2^0$ and $\dot{\epsilon}_3^0$ lie in the plane of σ_2^0 and σ_3^0 . Coincidence in the sense that the directions of $\dot{\epsilon}_2^0$ and $\dot{\epsilon}_3^0$ can be arbitrarily labelled as directions of σ_2^0 and σ_3^0 is attained. Coincidence in this sense was obtained for the Tresca material by the pictorial description of the previous section.

In the coincidence proof, curves lying on a yield surface were generated by rotating two of the principal stress directions about the third; more generally, curves lying on the yield surface can be generated by arbitrary rotations of the principal stress directions. It is interesting to note that all such curves correspond to a single point of principal stress space.

For the Mises material, normality implies further that the components of the strain rate tensor are proportional to the components of the stress deviator tensor. This conclusion is of course not valid for the Tresca material.

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